Thermoelectric Study of La$_2$Ti$_{2-x}$Nb$_x$O$_7$ (0≤x≤0.25) Ceramic Materials

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Authors’ contributions

This work was carried out in collaboration among all authors. Author ACI designed the study, performed the statistical analysis, wrote the protocol, and wrote the first draft of the manuscript. Authors FUI and EON managed the analyses of the study. Authors TCA and CVA managed the literature searches. All authors read and approved the final manuscript.

ABSTRACT

La$_2$Ti$_{2-x}$Nb$_x$O$_7$ (x = 0.00, 0.05, 0.10, 0.15, 0.20, 0.25) powders were synthesised via solid state reaction method, followed by sintering at 1673 K in a reducing atmosphere of 5% H$_2$/N$_2$ gas. The crystal structure, microstructure and thermoelectric (TE) properties of the pure and Nb-doped La$_2$Ti$_2$O$_7$ ceramics were investigated. All compositions were single phase with porous microstructures consistent with their low experimental densities. Thermoelectric results of Nb-doped compositions showed improved properties in comparison to pure La$_2$Ti$_2$O$_7$, suggesting that cation doping has the potential to improve the thermoelectric properties. Generally, the TE results obtained are not suitable for thermoelectric applications. However, the high Seebeck coefficient (≥ 190 μV/K) and glass-like thermal conductivity (≤ 2.26 W/m.K) values achieved have opened a new window for exploring the thermoelectric potentials of La$_2$Ti$_2$O$_7$ and other related oxides.

Keywords: Solid state reaction; thermoelectric; seebeck coefficient; thermal conductivity; La$_2$Ti$_2$O$_7$. 

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1. INTRODUCTION

Thermoelectric generators (TEGs) are solid state devices that convert a heat flux directly into electrical power [1]. The performance of TEGs is controlled by two main factors: functionality and efficiency of the semiconducting materials (thermoelements) from which the generator is composed. The functionality is based on a heavily doped p-type and n-type semiconducting materials with a high carrier concentration, \(n\) \((n \sim 10^5-10^{21} \text{ cm}^{-3})\) [2] and a bandgap of \(\sim 10 \text{ K}\) for optimized ZT [3]. Applying the process method of LTO, TiO\(_2\) and Ta-doped \(\text{La}_2\text{Ti}_3\text{O}_7\), respectivelly, at 573 K in pure \(\text{La}_2\text{Ti}_3\text{O}_7\), and very low values of \(\sim 1.12\) and \(\sim 0.93 \text{ W/m.K}\) for Sr- and Ta-doped \(\text{La}_2\text{Ti}_3\text{O}_7\), respectively at 573 K. These low \(k\) values are attributed to its large unit cells, large atomic mass, crystal anisotropy and complex crystal structure. As a complex structural material, it shows a high flexibility for tuning through cation/anion substitution which together with intrinsic low \(k\) are vital constituents required for a thermoelectric material.

The thermoelectric properties of LTO and other related layered perovskites have received less attention. Recently, Kholiq et al. [14] reported a low thermal conductivity value \((k \sim 1.3 \text{ W/m.K})\) at 573 K in pure \(\text{La}_2\text{Ti}_3\text{O}_7\), and very low values of \(\sim 1.12\) and \(\sim 0.93 \text{ W/m.K}\) for Sr- and Ta-doped \(\text{La}_2\text{Ti}_3\text{O}_7\), respectively at 573 K. These low \(k\) values are attributed to its large unit cells, large atomic mass, crystal anisotropy and complex crystal structure. As a complex structural material, it shows a high flexibility for tuning through cation/anion substitution which together with intrinsic low \(k\) are vital constituents required for a thermoelectric material.

\[ ZT = \frac{S^2 \sigma T}{k} \]  

Where \(S\) is the Seebeck coefficient, \(\sigma\) is the electrical conductivity, \(T\) is the absolute temperature and \(k = k_1 + k_2\) [4] is the total thermal conductivity. \(k_1\) and \(k_2\) are lattice thermal conductivity (thermal contributions from lattice vibrations/phonons) and electronic thermal conductivity (thermal contributions from charge carriers), respectively.

Non-oxide alloys such as Bi\(_2\)Te\(_3\)/Se\(_3\), Sb\(_2\)Te\(_3\), PbTe/Se and SiGe [5] presently are the most studied thermoelectric semiconducting materials. Due to their framework structures consisting of large voids with heavy elements, they possess low phonon group velocity and thus low \(k\) which are desirable for optimised ZT [6]. Application of these materials is limited, however, due to toxicity, scarcity, cost and limited operational temperature range [7]. According to Ioffe's theory, oxides are unsuitable for TE applications owing to their strong, mixed ionic and covalent bonds, high \(k_1\) and lower carrier mobilities resulting in low electrical conductivity [8]. Conversely, there is strong evidence that oxide thermoelectrics containing transition-metal-oxides are novel alternative materials to the conventional TE materials. Oxides are inert, nontoxic, light weight, cheap, possess small thermal expansion with high melting temperature, hence promising TE candidates for high temperature applications [9-11].

The study of electrical properties of lanthanum dititanate, \(\text{La}_2\text{Ti}_2\text{O}_7\) (LTO) ferroelectrics started over thirty years ago [12]. LTO belongs to the perovskite-like layered structure family with a homologous series \(\text{A}_x\text{B}_3\text{O}_{3n+2}\), where \(A\) = rare earth (RE) elements, \(B\) = titanium and \(n\) (= 4 in this case of LTO) is the number of octahedral units in the perovskite layers [13,14]. At room temperature, LTO possesses a monoclinic unit cell with a space group of P21 and a corresponding lattice parameter \(a = 13.0150 \text{ Å}, b = 5.5456 \text{ Å}, c = 7.8170 \text{ Å} and \beta = 98.6^\circ\) [14]. Above room temperature, LTO undergoes a transformation. For instance, at \(\sim 780^\circ\), it transforms to orthorhombic phase (CMc21) and changes to a paraelectric phase (CMcm) at \(1500^\circ\) C [12,15]. \(\text{La}_2\text{Ti}_2\text{O}_7\) exhibits high curie temperature, \(T_c > 1500^\circ\) C [12,13,15], excellent piezoelectric properties [12,13,16], non-linear optical and photocatalytic properties [12,13,15,16] and finds application in gas turbines at high temperature (> 1000\(^\circ\)C) [15].

The structure of LTO consists of discontinuous layers of corner sharing TiO\(_6\) octahedra in the unit cell [13,14] with La occupying interstitial positions between the octahedral Fig. 1 [13]. Pure LTO has a wide bandgap \((E_g)\) of 3 - 4 eV [17,18]. The size of \(E_g\) is dependent on the microstructure [17], morphology [18] and processing method of \(\text{La}_2\text{Ti}_2\text{O}_7\) [18].

Ceramic compositions of \(\text{La}_2\text{Ti}_2\text{O}_7\) \((0 \leq x \leq 0.25)\) were prepared by solid state reaction (SSR) method from the following starting powders: \(\text{La}_2\text{O}_3\) (99.99%, Sigma-Aldrich, UK), TiO\(_2\) (99.90%, Sigma-Aldrich, UK) and Nb\(_2\)O\(_5\) (99.50 %, Stanford Materials Corporation, USA). The raw powders \((\text{La}_2\text{O}_3, \text{TiO}_2\) and Nb\(_2\)O\(_5\)) were dried in a clean chamber furnace at 900°C for 8 h and cooled to 200°C. All dried powders were transferred to a vacuum desiccator and allowed to cool to room temperature in order to prevent adsorption of moisture. Stoichiometric proportions of the dried powders were mixed by ball milling in isopropanol with yttria stabilised zirconia (YSZ) milling media for 24 h. The slurry...
was dried at 80 °C, sieved through a 250 μm mesh and calcined at 1573 K in air for 6 h using alumina crucible [19]. The calcined powders were ground and pressed into a disc (20 mm diameter, ≤ 2 mm thickness) pellet in a uniaxial press. The green pellets were sintered in flowing 5% H₂/N₂ gas at 1673 K for 6 h.

Crystal structures of the ceramics were characterised by powder x-ray diffraction (XRD) with Cu Kα radiation (λ = 1.5406 Å) using D2 Phaser diffractometer (Bruker AX GmbH, Germany). The scan was conducted across the 2θ range of 20-60 degrees with a step size of 0.01° at a scan rate of 1°/min. Phase identity and purity of the collected data were analysed using the ICDD Sleve+PDF software [20]. Prior to microstructural examination, samples were ground, polished, thermally etched and carbon coated. Microstructures of the samples were studied using Philips XL 30 S-FEG scanning electron microscope.

Seebeck coefficient and electrical conductivity of disc samples were measured simultaneously in the temperature range of 573-973 K in an argon gas atmosphere using a Netzch SBA 458 Nemesis Seebeck and electrical conductivity analyser (Anter Flashline TM 3000) with a high speed xenon discharge (HSXSD) pulse source [21]. The experimental density of the ceramics was determined by Archimedes method using an electronic digital density balance (Mettler-Toledo AG balance).

3. RESULTS AND DISCUSSION

The room temperature XRD patterns of La₂Ti₂⁻ₓNbₓO₇ (LTO) ceramics are shown in Fig. 2(a). The patterns are indexed to a La₂Ti₂O₇ monoclinic structure ceramic (space group, P2₁) with the lattice constant a = 7.80896(10) Å, b = 5.54608(7) Å, and c = 13.01425(22) Å, [22] consistent with those reported in the literature [14,15]. No secondary phases were detected in any of the compositions within the detection limit of the diffractometer, and the peaks were sharp, suggesting a large particle size according to the Scherrer formula [14]. The bulk and relative density of all compositions for different Nb concentrations are plotted in Fig. 2(b). The relative density of all compositions varied progressively from 84 % to 93 %. This implies the bulk density increased with increase in Nb concentration with x = 0.25 showing the maximum sintering density of 5.38 g/cm³ (93 % of theoretical density, 5.789 g/cm³) [22].

The SEM images of the 5%H₂/N₂ sintered, thermally etched and carbon coated surfaces for La₂Ti₂⁻ₓNbₓO₇; 0.00 ≤ x ≤ 0.25 ceramics are shown in Fig. 3. The SEM images revealed homogenous and porous structures consistent with their low relative density of ≤ 93 % and average grain size of ≤ 2 μm. The effect of porosity on the thermoelectric performance of these compositions is unclear. However, some authors have suggested the presence of porosity in the lattice creates discontinuities which act as scattering centres thereby restricting carrier mobility and enhancing phonon scattering [21, 23,24]. As a result, both electrical conductivity, σ and thermal conductivity, k is reduced.

Fig. 4 and 5 show the temperature dependence of the electrical conductivity (σ), absolute Seebeck coefficient (|S|), power factor (PF), respectively for La₂Ti₂⁻ₓNbₓO₇ ceramic compositions. x = 0.00 (undoped La₂Ti₂O₇) exhibited the lowest σ in all the measured temperature range, consistent with its lowest density Fig. 2(b). The low σ obtained in x = 0.00 showed that carrier mobility was restricted probably by the inherent heavy pores in the grains. The electrical conductivity increased with
Fig. 2. XRD patterns of La$_2$Ti$_{2x}$Nb$_x$O$_7$ (LTO) ceramics (a) and the Archimedes measured density (b). The uncertainty in density measurement is ± 0.05%.

Fig. 3. SEM micrographs of the surfaces for La$_2$Ti$_{2x}$Nb$_x$O$_7$ (0.00 ≤ x ≤ 0.25) ceramics sintered 6 hours in 5% H$_2$/N$_2$ at 1773 K.

Nb doping but inconsistent with dopant (Nb) concentration. x = 0.10 showed the highest σ in all the measured temperature range Fig. 4(a), reaching a maximum of ~ 2.0 S/cm (200 S/m) at
873 K. This increase in electrical conductivity is attributed to the increase in carrier (electron) concentration due to the substitution of Nb$^{5+}$ for Ti$^{4+}$ which produces electrons. Moreover, oxygen vacancy, $V_O$ introduced by processing in reducing atmosphere increases the carrier concentration, thereby increasing $\sigma$ [25,26]. Some authors have also suggested the effect of grain size as a contributory factor to the enhanced $\sigma$. Doping has generally been observed to increase grain size, resulting in a reduced grain boundary area and scattering which may enhance the conduction [25-27]. The highest $\sigma$ (200 S/m) at 873 K for La$_2$Ti$_{1-x}$Nb$_x$O$_7$ obtained in this study is higher than the maximum $\sigma$ (0.5 S/m) reported in the literature for La$_{1+x}$Sr$_{0.4}$Ti$_2$O$_{5.815}$ ceramic at 573 K [14].

Fig. 4(b) shows the Seebeck coefficient, $|S|$ of La$_2$Ti$_{1-x}$Nb$_x$O$_7$ sample as a function of temperature. $S$ of all ceramics are negative, indicating that electrons are the dominant carriers [26,28-30]. $S$ increased monotonically with increasing temperature in all the measured temperature range. However, the behaviour of $S$ of the sample is inconsistent with Ioffe theory [31] (except $x = 0.00$ at 973 K). The relationship between $S$ and carrier concentration is given by the following equation [28]:

$$S = Y + \ln\frac{1}{n}$$

(2)

Where $Y$ and $n$ are the scattering factor and the carrier concentration, respectively. $S$ is inversely proportional to the carrier concentration. This implies that $x = 0.00$ with the lowest $\sigma$ (lowest carrier concentration) is expected to show the highest $S$ in all temperatures while $x = 0.10$ should likewise exhibit the lowest $S$ as a result its high $\sigma$ in obedience with Ioffe’s theory. At the maximum measured temperature (973 K), $x = 0.00$ as expected exhibited the highest absolute Seebeck coefficient value of $\sim$ 389 $\mu$V/K. This value is larger than values obtainable in most doped SrTiO$_3$ ceramics in the literature [28,32-35].

Combining the electrical conductivity and the Seebeck coefficient, the power factor (PF) of La$_2$Ti$_{1-x}$Nb$_x$O$_7$ sample was determined and shown in Fig. 5 as a function of temperature. Despite the high Seebeck coefficients (190 - 389 $\mu$V/K) exhibited by all the compositions, the PF remained very low (< 20 $\mu$W/K$^2$.m), due to the low $\sigma$ ($\leq$ 2.0 S/cm). However, the results obtained showed that the power factors of the Nb-doped compositions ($0.05 \leq x \leq 0.25$) where higher than that of undoped composition ($x = 0.00$) in all the measured temperature range, due to the enhanced electrical conductivity. $x = 0.10$ showed a higher PF value especially at high temperatures (773-973 K) than other compositions and recorded the maximum PF value of $\sim$ 18 $\mu$W/K$^2$.m at 973 K.

The temperature dependence of the total thermal conductivity, $k$ and the electronic thermal conductivity, $k_e$ of all samples are shown in Fig. 6. The thermal transport behaviour particularly $k$ of the Nb-doped ceramics is irregular with temperature. This behaviour could be related to the complex interplay of phonon scattering including U and N-processes on the ceramic material. Since cation doping of a material increases the grain size thereby promoting carrier (electron) mobility, it could be assumed that phonon propagation as well occurs. As a result, the Nb-doped La$_2$Ti$_2$O$_7$ ceramics exhibited a metallic conduction behaviour, which is evidenced in the increased $\sigma$ and $k$, respectively.

On the other hand, undoped La$_2$Ti$_2$O$_7$ showed the lowest $k$ across the measured temperature range, with a minimum $= 1.18$ W/m. K at 773-873 K. The reduced relative density observed in La$_2$Ti$_2$O$_7$ ceramics indicates an increase in porosity in the microstructure, which significantly affected thermal conductivity. The relation between the $k$ and volume of pores is given in the following equation [36]:

$$k = k_0 \left(1 - P^2\right)$$

(3)

Where $k_0$ is the thermal conductivity of the material without porosity and $P$ is the fraction of pores in the material. The implication of equation 3 therefore, is that increase porosity leads to an increase in phonon scattering, resulting in reduction of $k$. The minimal $k$ value (1.18 W/m.K) obtained in an undoped La$_2$Ti$_2$O$_7$ is lower than the value (1.3 W/m.K at 573 K) obtained in the literature for pure La$_2$Ti$_2$O$_7$ [14] and comparable to related polycrystalline PLS compounds such as Bi$_2$Ti$_2$O$_7$ (k $\sim$ 1W/m.K) [14, 37] and Sr$_2$Nb$_2$O$_7$ (k $\sim$ 1.5 W/m.K) [14,38]. For the Nb-doped La$_2$Ti$_2$O$_7$ compositions reported in this study, $x = 0.10$ showed the highest $k$ value (2.26 W/m. K) at 973 K, while $x = 0.05$ exhibited the lowest $k$ value of 1.49 W/m. K at 773 K, attributed to its large unit cell, large atomic mass, crystal anisotropy and complex crystal structure [37,38].
Fig. 4. Temperature dependence of electrical conductivity (a) and Seebeck coefficient (b) for pure and Nb-doped La$_2$Ti$_2$O$_7$ ceramics sintered 6 hours in 5% H$_2$/N$_2$ at 1773 K

Fig. 5. Temperature dependence of power factor for pure and Nb-doped La$_2$Ti$_2$O$_7$ ceramics sintered 6 hours in 5% H$_2$/N$_2$ at 1773 K

The electronic thermal conductivity of all LTO compositions showed similar temperature dependence with $\sigma$ and increased with increase in temperature as presented in Fig. 6(b).
From the small $k_E$ values ($\leq 0.0044$ W/m K), it is obvious to state that electronic thermal conductivity makes a very small contribution to the total thermal conductivity. This means that $k$ comes mainly from their lattice thermal conductivity [14,39].

The cumulative impact of the electrical and thermal transport properties on the thermoelectric performance is illustrated by the temperature dependence of the dimensionless figure of merit, ZT as shown in Fig. 7. From 573 K up to 873 K (573-873 K), $x = 0.00$ (pure La$_2$Ti$_2$O$_7$) had the lowest ZT values ($\leq 0.0045$) because of the low electrical conductivity. The combination of a higher $\sigma$ and $S$ resulted in a relatively high ZT in Nb-doped La$_2$Ti$_2$O$_7$ ceramics compared to undoped La$_2$Ti$_2$O$_7$. This suggests that the ZT of La$_2$Ti$_2$O$_7$ could be increased by a careful tuning with an appropriate dopant such as Nb. The ZT of all the compositions except $x = 0.15$ increased with increasing temperature within the measured temperature range. The ZT of La$_2$Ti$_{1.85}$Nb$_{0.15}$O$_7$ ($x = 0.15$) ceramic increased with temperature up to 873 K and decreased at 973 K. Furthermore, in all compositions, $x = 0.05$ and 0.10 showed a high ZT values at low temperatures (573-673 K), and beyond 673 K, $x = 0.05$ exhibited the highest values with a maximum ZT of 0.0084 at 973 K. The highest ZT displayed by 5 mol% Nb-doped La$_2$Ti$_2$O$_7$ is traceable to the lowest $k$ value recorded at high temperatures (773-973 K) compared to other Nb-doped La$_2$Ti$_2$O$_7$ ceramics.

Fig. 6. Temperature dependence of (a) total thermal conductivity (b) electronic thermal conductivity for pure and Nb-doped La$_2$Ti$_2$O$_7$ ceramics sintered 6 hours in 5% H$_2$/N$_2$ at 1773 K
4. CONCLUSIONS

Reduced Nb-doped La$_2$Ti$_2$O$_7$ ceramics were studied. The pure La$_2$Ti$_2$O$_7$ ceramics exhibited the lowest k (1.18 W/m.K) at 773 – 873 K and the maximum Seebeck coefficient, S (389 μV/K at 973 K). All doped compositions showed an increase in σ and k, attributed to metallic behaviour due to an increase in carrier concentration and creation of oxygen vacancies. x = 0.10 showed the highest PF (18 μW/K\textsuperscript{2}.m) at 973 K, resulting from its high electrical conductivity (σ\textsubscript{max} = 0.2 S/cm at 873 K) and moderate S. The highest ZT (0.0084) was recorded in x = 0.05 compositions at 973 K partially due to its low k value. Generally, Nb-doped La$_2$Ti$_2$O$_7$ ceramics exhibited very low TE properties especially σ, PF and ZT which are not suitable for thermoelectric applications. However, the high S (≥ 190 μV/K) and low k (≤ 2.26 W/m.K) recorded is a manifestation of the TE potential of Nb-doped La$_2$Ti$_2$O$_7$ ceramics.

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COMPETING INTERESTS

Authors have declared that no competing interests exist.

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